



INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

(51) International Patent Classification ⁶ : C30B 23/00, 29/16, 29/60, B32B 5/02, H01L 39/02	A1	(11) International Publication Number: WO 97/31139 (43) International Publication Date: 28 August 1997 (28.08.97)
---	-----------	---

(21) International Application Number: PCT/US97/02600

(22) International Filing Date: 21 February 1997 (21.02.97)

(30) Priority Data:

08/606,892	26 February 1996 (26.02.96)	US
08/790,824	22 January 1997 (22.01.97)	US

(71) Applicant: PRESIDENT AND FELLOWS OF HARVARD COLLEGE [US/US]; 17 Quincy Street, Cambridge, MA 02138 (US).

(72) Inventors: LIEBER, Charles, M.; 27 Hayes Avenue, Lexington, MA 02173 (US). YANG, Peidong; 339 Washington Street, 3, Somerville, MA 02143 (US).

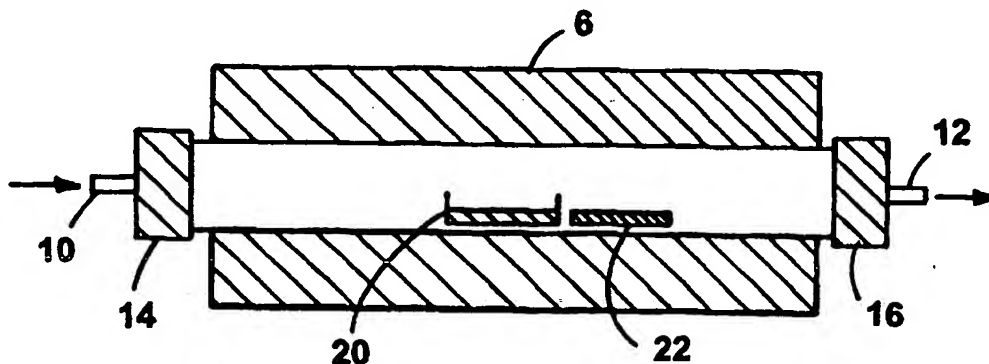
(74) Agents: TSAO, Y., Rocky et al.; Fish & Richardson P.C., 225 Franklin Street, Boston, MA 02110-2804 (US).

(81) Designated States: AU, CA, JP, European patent (AT, BE, CH, DE, DK, ES, FI, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE).

Published

*With international search report.**Before the expiration of the time limit for amending the claims and to be republished in the event of the receipt of amendments.*

(54) Title: METAL OXIDE NANORODS



(57) Abstract

This invention relates to metal oxide nanorods and composite materials containing such nanorods. The metal oxide nanorods have diameters between 1 and 200 nm and aspect ratios between 5 and 2000. The methods of producing the metal oxide nanorods include the steps of generating a metal vapor from a metal vapor source (20) in a furnace (6), exposing the nanorod growth substrate (22) to the metal vapor within a growth zone in the furnace (6) for a sufficient time to grow metal oxide nanorods on a surface of the nanorod growth substrate (22), removing the nanorod growth substrate (22) from the growth zone after the sufficient time to grow metal oxide nanorods on a surface of the nanorod growth substrate (22), and removing the metal oxide nanorods from the furnace (6). The methods can be used to prepare large quantities of metal oxide nanorods.

FOR THE PURPOSES OF INFORMATION ONLY

Codes used to identify States party to the PCT on the front pages of pamphlets publishing international applications under the PCT.

AM	Armenia	GB	United Kingdom	MW	Malawi
AT	Austria	GE	Georgia	MX	Mexico
AU	Australia	GN	Guinea	NE	Niger
BB	Barbados	GR	Greece	NL	Netherlands
BE	Belgium	HU	Hungary	NO	Norway
BF	Burkina Faso	IE	Ireland	NZ	New Zealand
BG	Bulgaria	IT	Italy	PL	Poland
BJ	Benin	JP	Japan	PT	Portugal
BR	Brazil	KE	Kenya	RO	Romania
BY	Belarus	KG	Kyrgyzstan	RU	Russian Federation
CA	Canada	KP	Democratic People's Republic of Korea	SD	Sudan
CF	Central African Republic	KR	Republic of Korea	SE	Sweden
CG	Congo	KZ	Kazakhstan	SG	Singapore
CH	Switzerland	LI	Liechtenstein	SI	Slovenia
CI	Côte d'Ivoire	LK	Sri Lanka	SK	Slovakia
CM	Cameroon	LR	Liberia	SN	Senegal
CN	China	LT	Lithuania	SZ	Swaziland
CS	Czechoslovakia	LU	Luxembourg	TD	Chad
CZ	Czech Republic	LV	Latvia	TG	Togo
DE	Germany	MC	Monaco	TJ	Tajikistan
DK	Denmark	MD	Republic of Moldova	TT	Trinidad and Tobago
EE	Estonia	MG	Madagascar	UA	Ukraine
ES	Spain	ML	Mali	UG	Uganda
FI	Finland	MN	Mongolia	US	United States of America
FR	France	MR	Mauritania	UZ	Uzbekistan
GA	Gabon			VN	Viet Nam

- 1 -

METAL OXIDE NANORODS

Background of the Invention

5 The invention relates to metal oxide nanoscale materials and composite materials based on the metal oxide nanoscale materials.

 Metal oxide whiskers are rod-shaped materials having diameters typically greater than or equal to 1 μm
10 are used extensively in composites due to their high strength and tolerance of high temperatures. Metal oxide whiskers in the same diameter range have been produced by a variety of procedures. Materials with nanometer diameters are useful in particular material applications,
15 such as magnetic information storage and pinning structures in superconductors. Quantum phenomena are expected from materials in the nanometer size regime. Metal carbide nanomaterials, some being rod-shaped, with diameters between 2 and 100 nm have been reported.

20 Summary of the Invention

 The invention relates to metal oxide nanorods, composite materials containing metal oxide nanorods, and methods of preparing metal oxide nanorods. The metal oxide nanorods have diameters between 1 and 200 nm and
25 aspect ratios between 5 and 2000.

 The metal oxide nanorods, including nanorods of MgO , Al_2O_3 , and ZnO , can be produced by controlled vapor-solid growth processes using a metal vapor source, such as a mixture of a bulk metal oxide powder and carbon
30 powder, and a low concentration of oxygen gas. Both bulk nanorods and nanorods aligned relative to substrate surfaces can be produced. Because the size regime for these metal oxide nanoparticles is smaller than that of

- 2 -

traditional whiskers, we term the materials nanorods. The reduction in size leads to materials with improved properties. For example, it is known that strength increases with decreasing diameter in this class of materials.

In one aspect, the invention relates to a metal oxide nanorod where the metal component contains a metal (e.g., one or more) selected from the group consisting of titanium, zirconium, hafnium, vanadium, niobium, tantalum, chromium, molybdenum, tungsten, manganese, technetium, rhenium, iron, osmium, cobalt, nickel, copper, zinc, scandium, yttrium, lanthanum, a lanthanide series element (e.g., cerium, praseodymium, neodymium, promethium, samarium, europium, gadolinium, terbium, dysprosium, holmium, erbium, thulium, ytterbium, and lutetium), boron, aluminum, gallium, indium, thallium, silicon, germanium, tin, lead, magnesium, calcium, strontium, and barium. The metal oxide nanorod has a narrow dimension (diameter) and a long dimension (length). Since the particles are nanorods, the narrow dimension of the particle is between 1 and 200 nm, more preferably between 2 and 50 nm. The nanorod can be further described by the aspect ratio (the ratio of the length to the diameter), which is preferably between 5 and 2000. Lengths are generally greater than 1 micrometer.

In examples, the metal oxide nanorod contains a metal component selected from the group consisting of magnesium, aluminum, zinc, cadmium, iron, and nickel. In other examples, the metal oxide nanorod is aligned relative to the surface of a substrate or a single crystal substrate. In certain examples, the nanorods are perpendicular to the surface, meaning that the nanorods are aligned within 15° of the substrate surface normal.

- 3 -

The metal oxide nanorod can be contained in a matrix material such as organic polymers, glass, ceramics, metals, or mixtures thereof. For example, the metal matrix may be a Ni/Al alloy or MoSi_2 . In examples, 5 the matrix material is a superconductor such as low temperature superconducting materials (e.g., Nb, NbTi, Nb_3Sn , or amorphous MoGe) or a high temperature copper oxide superconductor (e.g., $-\text{Bi}_2\text{Sr}_2\text{CaCu}_2\text{O}_{8+x}$ (BSCCO-2212), $\text{Bi}_2\text{Sr}_2\text{Ca}_2\text{Cu}_3\text{O}_{10+x}$ (BSCCO-2223), $\text{YBa}_2\text{Cu}_3\text{O}_{7+x}$, 10 $\text{Tl}_2\text{Ba}_2\text{Ca}_2\text{Cu}_3\text{O}_{10+x}$ (Tl-2223), $\text{Tl}_2\text{Ba}_2\text{CaCu}_2\text{O}_{8+x}$ (Tl-2212), $\text{TlBa}_2\text{CaCu}_2\text{O}_z$ (Tl-1212), $\text{TlBa}_2\text{Ca}_2\text{Cu}_3\text{O}_z$ (Tl-1223), $\text{HgBa}_2\text{CaCu}_2\text{O}_{6+x}$ (Hg-1212), or $\text{HgBa}_2\text{Ca}_2\text{Cu}_3\text{O}_{8+x}$ (Hg-1223)). In a composite formed on a substrate, the metal oxide nanorods can be aligned nearly perpendicular to the 15 surface of the substrate.

Another aspect of this invention relates to a method of producing metal oxide nanorods. The method involves the following three steps. First, a powdered metal vapor source is placed in a furnace (e.g., a tube 20 furnace), which can have a carrier gas flowing from one end of the furnace to the opposite end of the furnace. The carrier gas consists of an inert gas (e.g., argon, helium) and oxygen. The oxygen can be present as a trace amount in the furnace or the carrier gas. The preferred 25 oxygen concentration in the carrier gas is between 1 and 10 ppm. Second, a nanorod growth substrate is placed in the furnace, for example, downstream of the metal vapor source, within a 100°C temperature gradient of the metal vapor source, or within a growth zone. Finally, the 30 contents of the furnace are heated so that the temperature (e.g., the set point temperature) of the metal vapor source is between 500°C and 1600°C for up to two hours, generally between approximately 5 minutes and 2 hours. The nanorods can grow on the surface of the 35 nanorod growth substrate. The temperature is maintained

- 4 -

for a sufficient time to form metal oxide nanorods on a surface of the nanorod growth substrate. Growth of the nanorods can be halted by cooling the metal vapor source. Alternatively, the substrate can be removed from the growth zone after the growth of the metal oxide nanorods. The metal oxide nanorods are collected and removed from the furnace.

In another aspect, the invention features a method of producing metal oxide nanorods. First, a metal vapor is generated in a furnace, such as a tube furnace. Next, a nanorod growth substrate is exposed to the metal vapor within a growth zone in the furnace for a sufficient time to form metal oxide nanorods on a surface of the nanorod growth substrate. After the sufficient time to grow metal oxide nanorods on a surface of the nanorod growth substrate has passed, the nanorod growth substrate is removed from the growth zone. The metal oxide nanorods are collected and removed from the furnace. The nanorod growth substrate preferably can be a metal oxide nanocluster, a single crystal, graphite, or a metal coated substrate.

In yet another aspect, the invention features a method of producing magnesium oxide nanorods by placing a magnesium vapor source in a furnace, providing a nanorod growth substrate in the furnace, generating a magnesium vapor in the furnace by heating the magnesium vapor source (e.g., magnesium oxide and carbon powder) to a temperature between 500°C and 1600°C, exposing the nanorod growth substrate to the magnesium vapor within a growth zone in the furnace for a sufficient time to grow magnesium oxide nanorods on a surface of the nanorod growth substrate, removing the nanorod growth substrate from the growth zone after the exposing step, and collecting the magnesium oxide nanorods thus formed.

- 5 -

In examples, the nanorod growth substrate can be moved through the growth zone during the sufficient time to grow the metal oxide nanorods. The metal oxide nanorods can be purified by mechanically separating the metal oxide nanorods from the substrate (e.g., using a porous membrane). In examples where a graphite substrate is used, the metal oxide nanorods can be purified by heating the metal oxide nanorods in an oxidizing atmosphere to convert the graphite to carbon dioxide.

10 In examples, the metal vapor source includes an elemental metal or a mixture of metal oxide and carbon. The nanorod growth substrate, which is a substrate capable of supporting nanorod growth, contributes to the size of the nanocluster. The nanorod growth substrate may include metal oxide nanoclusters, a single crystal surface, graphite (e.g., carbon), a metal coated substrate (e.g., a substrate coated with a noble metal, such as gold, silver, copper, or platinum), a surface of the metal vapor source, or combinations thereof. The single crystal surface can be a film on a substrate or the surface of a bulk crystal. All of the nanorod growth substrates have the common feature of having or forming nanosized nucleation sites for growing the nanorods. The single crystal surface can be etched to form the nanosized nucleation sites or may form nanosized nucleation sites in situ. The nanorods may be encapsulated in a matrix material to form a composite.

As used herein, the term "nanorod" means a solid article having a narrow dimension (diameter) and a long dimension (length), where the ratio of the long dimension to the narrow dimension (the aspect ratio) is at least 5. In general, the aspect ratio is between 10 and 2000. By definition, the narrow dimension of the nanorod is a diameter between 1 and 200 nm. Accordingly, the length of the nanorod is between 0.01 and 400 μm . The nanorods

- 6 -

are solid structures composed, at least partially, of a metal oxide. The nanorods are single crystals. The term "narrow dimension," or alternatively "diameter," means the shortest dimension or cross-sectional thickness of a nanorod. Generally, the diameter of the nanorod is essentially constant along the length of the particular nanorod. The term "long dimension," or alternatively "length," means the longest dimension of the nanorod that is generally orthogonal to the diameter of the nanorod.

10 The phrase "nanosized nucleation site" means a site capable of initiating growth of a nanorod, such as on the surface of a nanorod growth substrate. Examples include etch pits in the surface of single crystals, metal coated substrates, nanoclusters, and combinations thereof. On graphite substrates, the nanosized nucleation site can form *in situ* under the reaction conditions. The phrase "aligned relative to a substrate" means that a majority (greater than 50%) of the nanorods extend in the same general direction from the surface of the substrate. For example, the (100) surface of a single crystal MgO substrate supports growth of MgO nanorods aligned perpendicular to the substrate surface. "Aligned perpendicular" means that the nanorods are within 15° of making a 90° angle with the substrate surface. The term "downstream" means that the substrate is positioned relative to the metal vapor source so that the carrier gas delivers metal vapor to the substrate. Generally, this means that the substrate is located beyond the metal vapor source in the direction that the carrier gas is flowing through the furnace. "Downstream" is further intended to describe that the substrate is in a lower temperature region of the furnace. There is a temperature gradient between the metal vapor source and the substrate.

- 7 -

As used herein, the "growth zone" is a region within a 100°C temperature drop between a location of highest temperature (or highest metal vapor concentration) and the location of the substrate. The
5 substrate is located within a 100°C temperature gradient of the metal vapor source.

The invention may provide one of the following advantages. The nanorods may have unique metallic, semiconducting, insulating, superconducting, optical, or
10 magnetic properties, or a combination thereof. Some embodiments have a lower density of stacking faults, as measured by TEM and normalized to diameter, than prior larger materials. The nanorods are highly anisotropic.

The metal oxide nanorods (e.g., magnesium oxide
15 nanorods) can be grown on a carbon or graphite substrate. The carbon substrate is relatively inexpensive and requires no pretreatment for nanorod growth, can be readily removed from the nanorods by oxidation to yield pure metal oxide nanorods, can lead to higher yield of
20 nanorods (per unit area of support), and can produce nanorods with larger aspect ratios relative to MgO etched single crystal substrates. In addition, the carbon substrates can be used in scaled-up commercial production of the metal oxide nanorods.

25 Other features and advantages of the present invention will be apparent from the following detailed description of the invention, and from the claims.

Brief Description of the Drawings

FIG. 1 is a schematic drawing of a batch reactor
30 for growth of metal oxide nanorods.

FIG. 2 is a schematic drawing of a quasi-continuous batch reactor for growth of metal oxide nanorods

- 8 -

FIG. 3 is a schematic drawing of a continuous flow reactor for growth of metal oxide nanorods.

Detailed Description

The invention relates to metal oxide nanorods, methods of preparing metal oxide nanorods, and composite materials containing metal oxide nanorods in a matrix material.

The nanorods are primarily composed of metal oxides of the general formula $M^1_x M^2_y O_z$, where M^1 and M^2 are metals selected from the group consisting of titanium, zirconium, hafnium, vanadium, niobium, tantalum, chromium, molybdenum, tungsten, manganese, technetium, rhenium, iron, osmium, cobalt, nickel, copper, zinc, scandium, yttrium, lanthanum, a lanthanide series element, boron, aluminum, gallium, indium, thallium, silicon, germanium, tin, lead, magnesium, calcium, strontium, and barium. The ratio of the metal to oxygen depends upon the oxidation state of the metal in the compound. In general, one of x and y, and z are integers greater than zero. That is to say, the compounds are stoichiometric and have few oxygen vacancies. When M^1 is the same as M^2 (or, alternatively, y is zero), the metal oxide is a binary metal oxide. Particular examples of binary metal oxides include MgO, Al_2O_3 , and ZnO.

Alternatively, the nanorods have a large number of oxygen vacancies. In effect, the materials are better described as metal doped metal oxides of the formula $M^2:M^1_x O_z$. The metal may be the same or different. In the case where M^1 and M^2 are the same, the material can be characterized as a metal oxide with an abundance or oxygen vacancies. The materials are metal doped metal oxides. Examples of compounds of this type include Zn:ZnO and In:ZnO. Compounds of this nature are formed,

- 9 -

for example, when carrier gases are less oxidizing than in cases where stoichiometric compositions are obtained.

The invention not only encompasses the metal oxide nanorods, but also other composite materials containing the nanorods. The dimensions of the nanorods permit the building of nanostructures and superior composites. For example, the tensile strength (kg/mm^2) of the nanorods is greater than that of the corresponding whisker. For example, bulk and film composites consisting of a high-temperature copper oxide superconductor matrix containing MgO nanorods can be prepared. The nanorods are analogous to columnar defects produced previously by ion irradiation, and are found to enhance significantly the critical current density and irreversibility temperature in these materials.

An alternative approach involves making columnar defect structures in materials by adding a pre-formed columnar structure to the superconducting matrix. There are several key criteria for making useful columnar defect structures. The rod-like structure must have a diameter of approximately 10 nm and a length greater than 0.1 μm . Additionally, the rod-like structure must not be chemically reactive with the superconductor matrix at processing temperatures.

25 Synthesis of metal oxide nanorods

The general synthetic approach used to prepare metal oxide nanorods involves three fundamental steps. First, a metal vapor must be generated from a metal source which is transported by a carrier gas to the substrate. Second, the metal vapor nucleates on the surface of the substrate. The substrate can be a pre-formed metal oxide nanocluster or single crystal defect site. Alternatively, the nucleation site can form *in situ* in the reactor, for example, when the substrate is carbon or graphite. This initiates growth

- 10 -

of the metal oxide nanorod. Finally, vapor phase growth of the nanorod takes place. The critical steps essential for the formation of nanorods are the nucleation and growth steps which define the initial size of the metal
5 oxide nanorod and its final diameter.

The metal source must be one that is a volatile source of metal atoms under the reaction conditions. One way to generate a metal vapor is by heating a source containing the elemental metal. Alternatively,
10 carbothermal reduction of bulk metal oxide powder can be used to generate volatile metal vapor. It is important that the carrier gas be composed of an inert gas for transporting the metal vapor to the substrate, and a small component of oxygen gas to make the deposition of
15 metal oxide possible. Commercially available reagents do not generally require pre-treatment before use as long as they are of sufficient purity (>99%). Commercial sources of starting materials and substrates include Aldrich Chemicals, Milwaukee, WI, Johnson-Mathey, Ward Hill, MA,
20 and Marketech International, Pittsburgh, PA.

The temperature profile for growing metal oxide nanorods can be determined from the phase diagram for the components of the nanorod or by experiment. In general, the substrate is placed downstream of the metal vapor
25 source. As a result, there is a temperature gradient between the metal vapor source and the substrate. Preferably, the substrate is located within a 100°C temperature gradient of the metal vapor source (where the concentration of metal vapor and temperature are
30 highest).

The properties of the nanorods can be determined, for example, by transmission electron microscopy (TEM) to measure crystallographic orientations and stacking fault density, energy-dispersive x-ray fluorescence to measure
35 the composition, x-ray diffraction (XRD) to measure

- 11 -

crystal structures, and convergent beam electron diffraction to measure lattice symmetry or the lattice constant a .

The metal oxide nanorods are ideally single
5 crystals and are primarily characterized by crystal structure. The single crystal materials consist of a single crystalline domain.

Without intending to be bound to any theories, it is believed the mechanism of nanorod formation involves
10 vapor-solid growth, where the nanoparticles or etched single crystal surfaces (or other substrates) provide nucleation sites for growth of the nanorods.

Alternatively, the nucleating surface, such as the carbon surface in single crystal graphite, has no preformed
15 nanosized nucleation sites. It is believed that the nucleation sites for nanorod growth form *in situ* either by the deposition of nanoclusters (e.g., MgO) from the vapor phase or by oxidation of the carbon growth surface which produces nanometer scale pits that can function as
20 nucleation sites. The average nanorod diameter is determined in part by the size of the nanoparticles during the bulk synthesis of the nanorods. In order to obtain high densities of aligned nanorods on the surface of a single crystal substrate, it is necessary to freshly
25 etch the surface of the single crystal substrate. This process provides a surface with numerous nanosized sites for nucleation and growth of nanorods, provided there is a good lattice match between the surface of the substrate and the growth plane of the nanorods.

30 Metal oxide nanorods can be prepared in a horizontal tube furnace batch reactor. Referring to FIG. 1, inert gas (e.g., argon or helium) flows through high-purity quartz tube furnace 6 via inlet 10 and outlet 12 located in end caps 14 and 16, respectively. The
35 inert gas flows over the metal vapor source 20 (e.g., MgO

- 12 -

+ carbon reactant in graphite boat). The metal oxide nanorods grow on nanorod growth substrate 22 (e.g., a carbon/graphite substrate).

Two approaches to large scale production of nanorods (e.g., magnesium oxide nanorods) utilize different reactor designs to achieve the enhanced production of the material. The reactor designs include: (1) a quasi-continuous batch-mode reactor; and (2) a continuous flow reactor. Both designs have been previously used in the large-scale growth of micron-scale whiskers of several different materials. See, for example, W.H. Sutton, "Principles and Methods for Fabricating Whisker-Reinforced Composite," in Whisker Technology, A.P. Levitt, Ed. (Wiley-Interscience, 1970, p. 318 and R.S. Wagner, "VLS Mechanism of Crystal Growth," in Whisker Technology p. 82.

In a quasi-continuous batch-mode reactor, metal vapor (e.g., magnesium) can be generated by the carbothermal reduction of a metal oxide (e.g., magnesium oxide) and transported to a nanorod growth substrate (e.g., graphite) in the growth zone of the reactor. Referring to FIG. 2, inert gas (e.g., argon or helium) flows through high-purity quartz tube furnace 6 via inlet 10 and outlet 12 located in end caps 14 and 16, respectively, and over the metal vapor source 20 (e.g., MgO + carbon reactant in graphite boat). The metal oxide nanorods grow on nanorod growth substrate 22 (e.g., a carbon/graphite substrate such as a graphite rod), which is mounted on manipulator 30. The nanorod growth substrate can be a substrate that has been seeded with preformed nanoclusters or, preferably, forms nanosized nucleation sites *in situ* (e.g., graphite). The metal oxide nanorods grow on a surface of the nanorod growth substrate. After the set growth time, the substrate is removed (e.g., retracted) from the growth zone using

- 13 -

manipulator 30. The resulting nanorods are removed from the substrate, for example, using scraper 40. The nanorods are collected for further use. The growth process can be continued by placing the nanorod growth substrate back into the growth zone of the reactor. It may be necessary to reseed the substrate before growing more nanorods on the surface. When a graphite substrate is used, it is not necessary to treat the substrate for nanorod growth.

10 In a continuous flow reactor, powdered nanorod growth substrate (e.g., preformed metal oxide nanoclusters or graphite powder) are admitted and allowed to pass through the growth zone where metal vapor (e.g., Mg(0)) reacts and produces nanorods that are collected

15 continuously at the bottom of the reactor. Referring to FIG. 3, inert gas (e.g., argon or helium) flows through vertical furnace 6 via inlet 10 and outlet 12. Inlet 10 is located in end caps 14 and outlet 12 is located at the opposite end of the furnace. The location of the metal

20 vapor source 20 (e.g., MgO + carbon reactant in graphite boat) in the furnace defines growth zone 21, where the growth of the nanorods is controlled. In the continuous process, the nanorod growth substrate 22 is a powder, such as graphite powder or metal oxide nanoclusters. The

25 powdered nanorod growth substrate is admitted from reservoir 24 through metering valve 26 into quartz tube in the a vertical tube furnace. The powdered substrate falls under the force of gravity through the growth zone 21 where the metal vapor (e.g., Mg(0)) is generated

30 (e.g., by the carbothermal reduction of magnesium oxide). The rate at which the substrate falls through growth zone 21 can be controlled by inert gas flow through inlet 10 and outlet 12. The falling rate can be used to regulate growth of the nanorods. The metering rate can also be

35 adjusted to alter the growth of the nanorods. The

- 14 -

resulting nanorods are collected continuously in collector 35 at the bottom of the reactor. The continuous flow reactor can be used to prepare multigram to kilogram quantities of metal oxide nanorods.

5 The metal oxide nanorods formed using graphite substrates can be purified, for example, by oxidizing graphite to carbon dioxide gas in an oxygen-containing atmosphere (e.g., pure oxygen or air) at a temperature above about 600°C. It is important that the temperature
10 be maintained below the sintering temperature of the metal oxide (e.g., 1800°C for magnesium oxide). In addition, it is possible that homogeneous nucleation may produce greater quantities of undesirable metal oxide clusters in the submicron size range in the flow reactor
15 than in the batch reactor. SEM and TEM can be used to identify the presence of metal oxide clusters, and conditions can be optimized to minimize the presence of these impurities. Furthermore, since the dimensions and flow characteristics of submicron clusters differ
20 significantly from the nanorods, it is possible to physically remove these impurities using controlled porosity membrane filters. Membrane filtration for purification and separation of particles has been described, for example, in W.H. Sutton, "Principles and
25 Methods for Fabricating Whisker-Reinforced Composite," in Whisker Technology, A.P. Levitt, Ed. (Wiley-Interscience, 1970, p. 315 and R. Lundberg, et al: "Processing of Whisker- reinforced ceramics," *Composites*, 18:125-127 (1987). Membrane filters can be used in large (kilogram)
30 scale separations, and are a straightforward method for further mechanically purifying the metal oxide nanorods.

Uses of metal oxide nanorods

 The metal oxide nanorods can be used in the preparation of nanostructures having superior mechanical,
35 electrical, optical and/or magnetic properties, such as

- 15 -

high density magnetic recording media. The metal oxide nanorods can be used as abrasives or environmentally responsive ("smart") materials. The small diameters and high aspect ratios of the nanorods make them useful in
5 metal, glass, organic polymer, and ceramic matrix composites. In particular, the metal oxide nanorods of the invention are useful as defects embedded within a superconductor.

For example, theoretical and experimental studies
10 have demonstrated that columnar defects enhance critical current densities and shift the irreversibility line to higher temperatures in copper oxide superconductors due to strong pinning of magnetic flux-lines in the superconductors. The columnar defects should have
15 diameters between 5 and 10 nm to maximize pinning while maintaining other desirable superconducting properties. Defects of these dimensions have been obtained by irradiating samples with high energy heavy ions or high energy protons. Pinning structures in superconductors
20 are described in, for example, P. LeDoussal and D.R. Nelson, *Physica C* 232:69-74 (1994).

Nanorod-superconductor composites may be useful materials for superconducting wires used in power transmission and magnetic solenoids or thin film devices
25 requiring large current densities and power throughputs such as send/receive filters in cellular communication base stations.

Metal oxide nanorod-superconductor composites

Metal oxide nanorod-superconductor composites can
30 be made in bulk and film form (e.g., superconducting tapes, wires, cables, or films, such as in single and multifilament wires fabricated using BSCCO-2212 or BSCCO-2223). The composites have a columnar defect structure and enhanced critical current behavior.
35 Composites can be prepared using MgO nanorods since these

- 16 -

materials appear to react the least with high temperature superconductors such as $\text{Bi}_2\text{Sr}_2\text{CaCu}_2\text{O}_8$ (BSCCO-2212) superconductors, although it can be applied to other high temperature superconductors and BSCCO phases by simple
5 modification of experimental conditions.

Bulk composite materials were made by combining nanorods and pre-reacted BSCCO powder. The resulting mixture is melt-textured on silver foil using procedures similar to making Ag/BSCCO tapes, as described in, for
10 example, K. Nomura, et al., *Applied Physics Letters*, 62:2131-2133 (1993). TEM structural analysis of these bulk composites demonstrate that MgO nanorods are incorporated into crystal grains of BSCCO, and that nanorods are oriented principally perpendicular and
15 parallel to the Cu-O planes of the superconductor.

Film composite materials were prepared by depositing amorphous BSCCO material at low temperature onto a MgO substrate containing aligned MgO nanorods with a rod density of 2×10^9 to $2 \times 10^{10} \text{ cm}^{-2}$. The composite
20 was subsequently melt-textured thus causing crystallization and alignment of the BSCCO matrix. X-ray diffraction shows that the melt-textured BSCCO is aligned with the c-axis perpendicular to the substrate surface. In addition, TEM analysis demonstrates clearly that MgO
25 nanorods pass through the BSCCO Cu-O planes, and have a size and density comparable to that of the starting substrate.

As expected, the critical current densities of the nanorod-BSCCO composites are enhanced dramatically
30 compared to reference samples. The enhancements are comparable to those obtained previously using heavy-ion irradiation techniques (See, e.g., P. Kummeth, et al., *J. Alloys and Compounds* 195:403-406).

Without further elaboration, it is believed that
35 one skilled in the art can, based on the description

- 17 -

herein, utilize the present invention to its fullest extent. All publications cited herein are hereby incorporated by reference. The following specific examples are, therefore, to be construed as merely
5 illustrative, and not limitive of the remainder of the disclosure.

Example 1. Bulk synthesis of MgO nanorods on nanoclusters.

MgO powder and graphite powder were mixed in a
10 weight ratio between 1:3 and 3:1. The resulting powder mixture was loaded into a graphite boat. The graphite boat was placed into a quartz tube flushing with Ar gas with a flow rate of between 100 and 2000 sccm. The gas contained between 1 and 10 ppm oxygen. The substrate was
15 placed in the tube downstream of the graphite boat where the temperature gradient between the graphite boat and the substrate was less than 100°C. In the synthesis of bulk MgO nanorods, the substrate was composed of MgO nanoclusters with average diameters between 5 and 50 nm.
20 The nanorods were obtained by heating the quartz tube to approximately 1200°C for 0.5 to 2 hours.

The nanorods of MgO grew on the surface of the nanoclusters. The nanorods have diameters in the range from 5-100 nm depending on the reaction time, initial
25 substrate nanocluster size and local supersaturation. The lengths of the nanorods ranges from 1-10 μ m. The reaction times, initial nanocluster diameters and nanorod dimensions are listed in Table I.

- 18 -

Table I

Nanocluster		Reaction	Nanorod	
<u>Diameter (nm)</u>		<u>Time (min)</u>	<u>Diameter (nm)</u>	<u>Length (μm)</u>
5 - 50		30	10 - 50	1 - 10
5	5 - 50	60	50 - 100	1 - 10
	30 - 150	60	50 - 150	1 - 10

Example 1a. Bulk synthesis of MgO nanorods on different substrates.

A mixture of MgO and graphite powder with a weight ratio of 1:3 to 3:1 was mixed by grinding, and placed in a graphite furnace boat. The graphite boat was placed into a quartz furnace tube and positioned at the center of a horizontal tube furnace. The tube containing the graphite boat and reactants was then flushed with argon gas at a flow rate between 100 and 2000 sccm. The substrate was placed downstream of the tube within -100°C of the boat. In runs numbered 9 through 12, the surface of the MgO/C mixture acted as the substrate (i.e., no other substrate was included in the furnace. After loading the furnace, it was heated to a set point temperature between 1000°C and 1200°C measured at the center point of the furnace. The center point of the furnace was maintained at the set point temperature for times ranging from zero minutes up to 60 minutes (i.e., the hold time), depending on the substrate, at which time the contents of the furnace were cooled back to room temperature.

Table Ia					
No.	Substrate	Set Point Temperature (°C)	Hold Time at Set Point (min)	Diameter (nm)	Length (μm)
1	Highly Oriented Pyrolytic Graphite	1200	0	5-90	1-5
2	Carbon Plate Graphite Sheet	1150	0	5-80	1-5
3	Carbon Plate Graphite Sheet	1200	0	20-150	1-10
4	Carbon Plate Graphite Sheet	1200	30	50-200	1-50
5	Au/Graphite Sheet	1125	0	5-100	1-5
6	Au/Graphite Sheet	1200	0	5-150	1-10
7	Au/MgO	1125	0	5-40	1-5
8	Au/Etched MgO	1125	0	5-40	1-5
9	Surface of MgO/C Mixture	1100	60	50-200	1-200
10	Surface of MgO/C Mixture	1125	0	5-100	1-5
11	Surface of MgO/C Mixture	1200	0	5-100	1-5
12	Surface of MgO/C Mixture	1200	60	50-200	1-200

15 The diameters and lengths of the nanorods were measured and are summarized along with types of substrates and reaction conditions in Table Ia. For example, the nanorods grown on the graphite plate have diameters ranging from 5 nm to 200 nm and have aspect

20 ratios typically of 50 to 1000. Smaller diameters can be favored by reducing the reaction time (e.g., to 0-10 minutes from 30 minutes). The packing density of nanorods on the surface of the gold coated etched magnesium oxide substrate (No. 8) was $2 \times 10^{10} \text{ cm}^{-2}$.

- 20 -

Example 2. MgO nanorods aligned on a substrate surface.

The nanorods were prepared by the methods described in Example 1, except in order to obtain 5 nanorods aligned on a substrate surface, a single crystal that has been freshly etched was used as the substrate. Specifically, the substrate was the (100) face of a MgO single crystal which was etched with 0.1 to 0.5 M aqueous NiCl_2 solution for 0.25 to 2 hours. The 10 substrate was placed in the furnace immediately after etching. The etching time effects the density of the nanorods on the surface. Typically, etching times of 0.5 hours and etching solution concentrations of 0.5 M are used.

15 The nanorods grown on the single crystal surface are relatively uniform in diameter and length. The diameter ranges from 10 to 30 nm the length varies from and 1 to 2 μm . The nanorods are oriented on the crystal surface. Most (e.g., greater than 50%, more preferably 20 greater than 80%) of the nanorods are perpendicular to the (100) surface while the others are at a 45° angle to the substrate surface. The packing density of the nanorods on the surface is between 1×10^7 and $5 \times 10^9 \text{ cm}^{-2}$ depending on the concentration of the etching solution 25 and the etching time. Particular etching times, etching solution concentrations and nanorod densities are listed in Table II.

Table II

	<u>Etching Time (min)</u>	<u>Density of nanorods (cm^{-2})</u>
30	0	5×10^8
	15	1×10^9
	30	5×10^9

Note: The etching solution concentration was 0.5 M.

- 21 -

Example 3. Bulk synthesis of Al_2O_3 nanorods.

Al_2O_3 powder and graphite powder were mixed in a weight ratio of between 1:3 and 3:1. The mixture was loaded into a graphite boat. The graphite boat was put
5 into a alumina tube flushing with Ar gas at a flow rate between 100 and 2000 sccm. The gas contained between 1 and 10 ppm oxygen. Al_2O_3 nanoclusters with average diameters between 5 and 10 nm were put downstream of the graphite boat where the temperature gradient between the
10 graphite boat and the substrate was 100°C. The tube was heated to 1400°C for 0.5 to 2 hr to generate the nanorods.

The nanorods of Al_2O_3 collected on the surface of nanoclusters. The nanorods have diameters ranging from 5
15 to 100 nm and lengths ranging from 1 to 10 μm .

Example 4. ZnO nanorods aligned on a substrate surface.

ZnO powder and graphite powder was mixed in a weight ratio 1:3 to 3:1. The mixture was loaded into a
20 graphite boat which was placed into a quartz tube being flushed with Ar gas with a flow rate between 100 and 2000 sccm. The gas contained between 1 and 10 ppm oxygen. A (100) SrTiO_3 single crystal substrate coated with a film of gold with a thickness between 1 and 100 nm was placed
25 in the tube downstream of the graphite boat where the temperature gradient between the graphite boat and the substrate was approximately 300°C. The quartz tube was heated to between 900°C and 1000°C for between 0.25 to 1 hour to grow the nanorods.

30 Alternatively, zinc metal can be used as the metal vapor source, in which case the tube furnace was heated to between 500°C and 700°C.

The nanorods grown on the single crystal surface were uniform in diameter and length, ranging from 10 to

- 22 -

50 nm and 1 to 2 μm , respectively. Nearly all of the nanorods were oriented relative to the crystal surface, but were not normal to the surface.

Example 5. Bulk nanorod-superconductor

5 composites.

MgO nanorods with diameters between 10 and 100 nm and lengths between 1 and 10 μm were mixed and ground with pre-reacted BSCCO(2212) powder. The mixture contained between 0 and 15 percent nanorods by weight.

10 The fine powder mixture was pressed into a pellet using a pellet compressor (e.g., 10 tons over a 20 mm diameter die). The resulting pellet was placed onto the surface of Ag foil and was heated to between 860 and 880°C. The material was maintained in this partially melted state

15 for between 10 and 30 minutes. The temperature was decreased at a rate of between 1 and 5°C per hour to between 760 and 800°C. The material was held at this temperature for between 10 and 24 hours at which time the sample was cooled down to room temperature to yield the

20 final composite. The composite had much higher critical current densities than a reference sample processed in a similar manner that did not contain the MgO nanorods. At 5K, 0 Tesla, the critical current density of the composite in comparison to a reference sample lacking

25 nanorods increased from 2×10^4 to $2 \times 10^5 \text{ A cm}^{-2}$, and at 30K, 0 Tesla, it increased from 4×10^3 to $5 \times 10^4 \text{ A cm}^{-2}$. Over the temperature range of 5 to 60K, there was a one magnitude increase in critical current density.

Example 6. Film nanorod-superconductor

30 composites.

The MgO nanorods were first grown on the (100) surface of a MgO single crystal substrate as described above in Example 2. Amorphous BSCCO(2212) was deposited

- 23 -

on the substrate using pulsed laser deposition at room temperature. Pulsed laser deposition is described in, for example, S. Zhu, et al., *Applied Physics Letters*, 63:409-411 (1993). The amorphous composite was

5 subsequently melt-textured using the heating program described for the bulk composite in Example 5 to afford the final film composite material. The critical current density of the film composite is much greater than that of a reference sample without nanorods. At 5K, 0.8

10 Tesla, the critical current density of the composite in comparison to a reference sample lacking nanorods increased from 2×10^4 to 1.5×10^5 A cm⁻², at 40K, 0.5 Tesla, it increased from 1×10^3 to 2×10^4 A cm⁻², and at 60K, 0.15 Tesla, it increased from 8×10^2 to 1×10^4

15 A cm⁻².

Example 7. MgO nanorod/Tl₂Ba₂Ca₂Cu₃O₁₀ composite.

The MgO nanorod array was first grown on the (100) surface of a MgO single crystal substrate as described above in Example 2. Amorphous Ba₂Ca₂Cu₃O₇ was deposited

20 on onto the MgO nanorod forest using pulsed laser deposition at 200°C. Then, the sample was thallinated and crystallized to form the Tl₂Ba₂Ca₂Cu₃O₁₀ superconductor in a three-zone furnace. During the thallination, the sample is sealed in Ag foil with a

25 small amount of Tl₂Ba₂Ca₂Cu₃O₁₀ powder and placed at one end of an alumina tube. The Tl source (Tl₂O₃) was placed at the other end of the alumina tube. The alumina tube was half sealed with Ag foil and placed into the three-zone furnace. The sample temperature was kept at

30 790°C and the Tl source temperature was maintained at 890°C. This thallination process is described in, for example, W.L. Holstein, et al., *J. Mater. Res.*, 11(6):1349 (1996). The critical current density of the final film composite was much greater than that of a

- 24 -

reference sample without nanorods. At 50K, 0.4 Tesla, the critical current density of the composite in comparison to a reference sample lacking nanorods increased from 1×10^3 to 2×10^4 A cm⁻², at 60K, 0.3 Tesla, it increased from 1×10^3 to 1.5×10^4 A cm⁻², and at 90K, 0.15 Tesla, it increased from 100 to 7000 A cm⁻².

Other Embodiments

From the above description, the essential characteristics of the present invention can be ascertained. Without departing from the spirit and scope thereof, various changes and modifications of the invention can be made to adapt it to various usages and conditions. Thus, other embodiments are also within the claims.

- 25 -

Claims

1. A metal oxide nanorod comprising a metal selected from the group consisting of titanium, zirconium, hafnium, vanadium, niobium, tantalum,
5 chromium, molybdenum, tungsten, manganese, technetium, rhenium, iron, osmium, cobalt, nickel, copper, zinc, scandium, yttrium, lanthanum, a lanthanide series element, boron, aluminum, gallium, indium, thallium, silicon, germanium, tin, lead, magnesium, calcium,
10 strontium, and barium; wherein said metal oxide nanorod has a narrow dimension between 1 and 200 nanometers, and an aspect ratio between 5 and 2000.
2. The metal oxide nanorod of claim 1, wherein said metal is selected from the group consisting of
15 magnesium, aluminum, zinc, cadmium, iron, and nickel.
3. The metal oxide nanorod of claim 2, wherein said narrow dimension is between 5 and 50 nanometers.
4. The metal oxide nanorod of claim 1, wherein said metal oxide nanorod is aligned relative to the
20 surface of a substrate.
5. The metal oxide nanorod of claim 1, further comprising a matrix material including a glass, ceramic, organic polymer, or metal; wherein said metal oxide nanorod is contained in said matrix material.
- 25 6. The metal oxide nanorod of claim 5, wherein said metal oxide nanorod is aligned perpendicular to the surface of a substrate.
7. The metal oxide nanorod of claim 6, wherein said matrix material is a superconductor.

- 26 -

8. The metal oxide nanorod of claim 7, wherein said superconductor is a high temperature copper oxide superconductor.

9. A nanorod composite comprising a metal oxide
5 nanorod contained in a surrounding matrix material, wherein

said metal oxide nanorod includes a metal selected from the group consisting of titanium, zirconium, hafnium, vanadium, niobium, tantalum, chromium,
10 molybdenum, tungsten, manganese, technetium, rhenium, iron, osmium, cobalt, nickel, copper, zinc, scandium, yttrium, lanthanum, a lanthanide series element, boron, aluminum, gallium, indium, thallium, silicon, germanium, tin, lead, magnesium, calcium, strontium, and barium; has
15 a narrow dimension between 1 and 200 nanometers; and has an aspect ratio between 5 and 2000; and

said matrix material includes glass, ceramic, organic polymer, or metal.

10. The nanorod composite of claim 9, wherein
20 said metal is selected from the group consisting of magnesium, aluminum, zinc, cadmium, iron, and nickel.

11. The nanorod composite of claim 10, wherein said matrix material is a superconductor.

12. The nanorod composite of claim 11, wherein
25 said matrix material is a high temperature copper oxide superconductor.

13. The nanorod composite of claim 12, wherein said metal oxide nanorods are aligned relative to the surface of a substrate.

- 27 -

14. The nanorod composite of claim 13, wherein said metal oxide nanorod is a magnesium oxide nanorod.

15. A method of producing metal oxide nanorods comprising:

- 5 generating a metal vapor in a furnace;
 exposing a nanorod growth substrate to the metal vapor within a growth zone in the furnace for a sufficient time to form metal oxide nanorods on a surface of the nanorod growth substrate;
- 10 removing the nanorod growth substrate from the growth zone; and
 collecting the metal oxide nanorods thus formed.

16. The method of claim 15, wherein the nanorod growth substrate is a metal oxide nanocluster, a single
15 crystal, graphite, or a metal coated substrate.

17. The method of claim 16, wherein the growth zone is a region within a 100°C temperature drop between a location of highest metal vapor concentration and a location of the substrate.

20 18. The method of claim 17, wherein the metal vapor source includes magnesium.

19. The method of claim 18, further comprising mechanically separating the metal oxide nanorods from the substrate.

25 20. The method of claim 19, wherein the nanorod growth substrate is graphite.

- 28 -

21. The method of claim 20, further comprising purifying the metal oxide nanorods by heating the metal oxide nanorods in an oxidizing atmosphere to convert the graphite to carbon dioxide.

5 22. The method of claim 15, further comprising the step of moving the nanorod growth substrate through the growth zone during the exposing step.

23. A method of producing metal oxide nanorods comprising:

10 placing a metal vapor source in a furnace;
 providing a nanorod growth substrate in the furnace;

 heating the metal vapor source to a temperature between 500°C and 1600°C;

15 maintaining the temperature for a sufficient time to form metal oxide nanorods on a surface of the nanorod growth substrate; and

 collecting the metal oxide nanorods thus formed.

24. The method of claim 23, further comprising
20 flowing a carrier gas through the furnace.

25. The method of claim 24, wherein said providing step includes placing the nanorod growth substrate downstream of the metal vapor source.

26. The method of claim 23, wherein said
25 providing step includes placing the nanorod growth substrate within a growth zone.

- 29 -

27. The method of claim 26, wherein the growth zone is a region within a 100°C temperature gradient drop between the metal vapor source and the substrate when heated to the temperature between 500°C and 1600°C.

5 28. The method of claim 27, further comprising removing the nanorod growth substrate from the growth zone after the sufficient time to form metal oxide nanorods on a surface of the nanorod growth substrate.

29. The method of claim 28, further comprising
10 mechanically separating the metal oxide nanorods from the substrate.

30. The method of claim 29, wherein the nanorod growth substrate is graphite.

31. The method of claim 30, further comprising
15 purifying the metal oxide nanorods by heating the metal oxide nanorods in an oxidizing atmosphere to convert the graphite to carbon dioxide.

32. The method of claim 31, wherein the metal oxide is magnesium oxide.

20 33. The method of claim 28, further comprising moving the nanorod growth substrate through the growth zone during the sufficient time to form the metal oxide nanorods.

34. The method of claim 23, further comprising
25 cooling the metal vapor source to stop the formation of the metal oxide nanorods.

- 30 -

35. The method of claim 23, wherein the metal vapor source includes an elemental metal or a mixture of metal oxide and carbon, wherein the metal is selected from the group consisting of titanium, zirconium, hafnium, vanadium, niobium, tantalum, chromium, molybdenum, tungsten, manganese, technetium, rhenium, iron, osmium, cobalt, nickel, copper, zinc, scandium, yttrium, lanthanum, a lanthanide series element, boron, aluminum, gallium, indium, thallium, silicon, germanium, tin, lead, magnesium, calcium, strontium, and barium.

36. The method of claim 23, wherein the nanorod growth substrate includes a metal oxide nanocluster, a surface of the metal vapor source, a single crystal, graphite, or a metal coated substrate.

37. The method of claim 36, wherein said metal oxide is magnesium oxide.

38. A method of producing magnesium oxide nanorods comprising:

placing a magnesium vapor source in a furnace;
providing a nanorod growth substrate in the furnace;
generating a magnesium vapor in the furnace by heating the magnesium vapor source to a temperature between 500°C and 1600°C;
exposing the nanorod growth substrate to the magnesium vapor within a growth zone in the furnace for a sufficient time to form magnesium oxide nanorods on a surface of the nanorod growth substrate, the growth zone being a region in the furnace within a 100°C temperature drop of the magnesium vapor source;
removing the nanorod growth substrate from the growth zone after the sufficient time to grow metal oxide

- 31 -

nanorods on a surface of the nanorod growth substrate;
and

collecting the magnesium oxide nanorods thus
formed.

5 39. The method of claim 38, wherein the nanorod
growth substrate is graphite, metal coated substrate, a
surface of the magnesium vapor source, or a magnesium
oxide nanocluster.

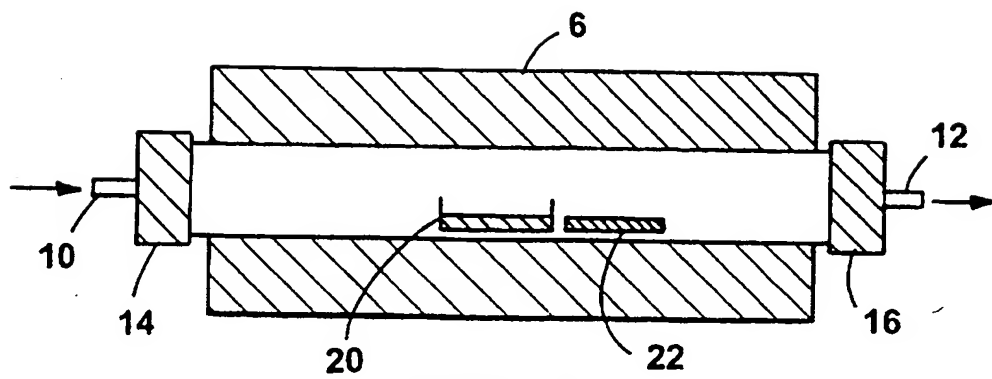
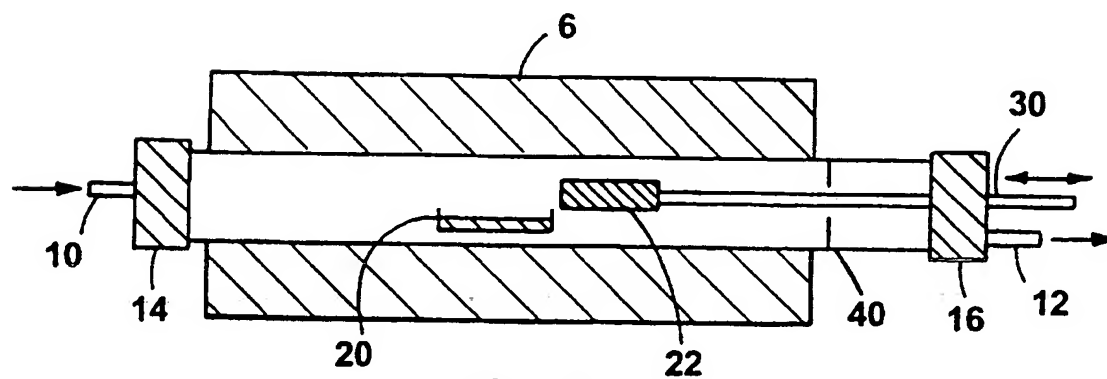
10 40. The method of claim 39, wherein the nanorod
growth substrate is graphite.

41. The method of claim 40, further comprising
purifying the metal oxide nanorods by heating the metal
oxide nanorods in an oxidizing atmosphere to convert the
graphite to carbon dioxide.

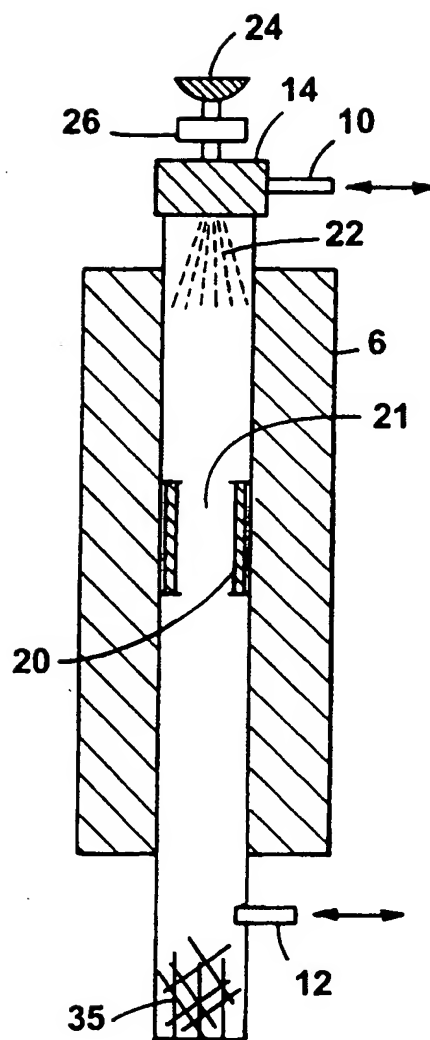
15 42. The method of claim 41, further comprising
moving the nanorod growth substrate through the growth
zone during the exposing step.

43. The method of claim 42, further comprising
mechanically separating the metal oxide nanorods from the
20 substrate.

1/2

**FIG. 1****FIG. 2**

2/2

**FIG. 3**

INTERNATIONAL SEARCH REPORT

International application No.
PCT/US97/02600

A. CLASSIFICATION OF SUBJECT MATTER

IPC(6) : C30B 23/00, 29/16, 29/60; B32B 05/02; H01L 39/02

US CL : Please See Extra Sheet.

According to International Patent Classification (IPC) or to both national classification and IPC

B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)

U.S. : Please See Extra Sheet.

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched
none

Electronic data base consulted during the international search (name of data base and, where practicable, search terms used)

Please See Extra Sheet.

C. DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
X	US 3,951,677 A (JACOBSON et al) 20 April 1976, col. 1, lines 27-59 and col. 4, lines 33-45.	1-3, 5, 9, 10
X	US 4,778,716 A (THORFINNSEN et al.) 18 October 1988, col. 2, lines 15-26 and col. 3, line 32 to col. 4, line 19.	1, 3, 5, 9
X	US 5,418,007 A (DEBE) 23 May 1995, col. 4, lines 19-26, col. 6, lines 4-14 and 23-27, col. 7, lines 18-52, col. 8, lines 9-13, and col. 11, line 66 to col. 12, line 15.	1-6, 9, 10
X	US 5,441,726 A (MITCHNIK et al.) 15 August 1995, col. 2, lines 22-67, col. 3, line 65 to col. 4, line 13.	1-5, 9, 10

☒ Further documents are listed in the continuation of Box C. ☐ See patent family annex.

•	Special categories of cited documents:	•T	later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention
•A	document defining the general state of the art which is not considered to be of particular relevance		
•E	earlier document published on or after the international filing date	•X	document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step when the document is taken alone
•L	document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified)	•Y	document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art
•O	document referring to an oral disclosure, use, exhibition or other means		
•P	document published prior to the international filing date but later than the priority date claimed	•A	document member of the same patent family

Date of the actual completion of the international search

15 MAY 1997

Date of mailing of the international search report

26 JUN 1997

Name and mailing address of the ISA/US
Commissioner of Patents and Trademarks
Box PCT
Washington, D.C. 20231

Facsimile No. (703) 305-3230

Authorized officer

MARIE K. YAMNITZKY

Telephone No. (703) 308-4413

INTERNATIONAL SEARCH REPORT

International application No.
PCT/US97/02600

C (Continuation). DOCUMENTS CONSIDERED TO BE RELEVANT

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
X,P	US 5,569,445 A (FUKATSU et al) 29 October 1996, col. 1, lines 8-15, col. 12, lines 20-36, and Tables 1-4.	1-5, 9, 10

INTERNATIONAL SEARCH REPORT

International application No.
PCT/US97/02600

A. CLASSIFICATION OF SUBJECT MATTER:

US CL :

428/401, 364, 699, 700, 701, 702, 328, 330, 332, 293.1, 293.4, 148; 117/87, 107; 505/230, 238

B. FIELDS SEARCHED

Minimum documentation searched

Classification System: U.S.

428/401, 364, 375, 699, 700, 701, 702, 323, 328, 329, 330, 332, 293.1, 293.4, 148, 149, 903, 930; 117/87, 107, 902, 921, 944; 505/230, 238

B. FIELDS SEARCHED

Electronic data bases consulted (Name of data base and where practicable terms used):

APS/USPAT file

search terms: nanorod?, nanowhisker?, nanofiber?, nanofibre?, nanometer#, nanometre#, nm, angstrom#, ang, fiber#, fibre#, whisker#, rod#, diameter#, aspect ratio#, oxide#, metal, silicon, single crystal?, superconduct?, super conduct?, mgo, magnesium oxide#